

U.S. PATENT APPLICATION

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Invention: HEAT TREATABLE COATED ARTICLE WITH TIN OXIDE INCLUSIVE
LAYER BETWEEN TITANIUM OXIDE AND SILICON NITRIDE

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SPECIFICATION

TITLE OF THE INVENTION

HEAT TREATABLE COATED ARTICLE WITH TIN OXIDE INCLUSIVE LAYER BETWEEN TITANIUM OXIDE AND SILICON NITRIDE

[0001] The instant application relates to a heat treatable coated article which includes a layer comprising tin oxide located between a layer comprising a metal oxide such as titanium oxide and a layer comprising a nitride such as silicon nitride. Surprisingly, it has been found that the provision of a tin oxide inclusive layer between such layers can significantly improve durability of the resulting coated article, especially after heat treatment (HT).

CROSS-REFERENCE TO RELATED APPLICATIONS

[0002] This application is related to each of U.S. Patent Application Serial Nos. 10/400,080, filed March 27, 2003, 10/636,918, filed August 8, 2003, 10/337,383, filed January 7, 2003, and 10/337,384, filed January 7, 2003 which is a division of 09/794,224, filed February 28, 2001 (now U.S. Patent No. 6,576,349), the disclosures of which are all hereby incorporated herein by reference.

BACKGROUND OF THE INVENTION

[0003] Windows including glass substrates with solar control coatings provided thereon are known in the art. Such windows may be used in the context of architectural windows, insulating glass (IG) window units, automotive windows such as windshields, and/or the like.

[0004] Related U.S. Patent Application Serial No. 10/400,080 discloses at least a coated article having the following layer stack where the layers are listed in order from the glass substrate outwardly: glass/TiO_x(40 Å)/Si_xN_y(113 Å)/ZnO_x(100 Å)/Ag(95 Å) and so forth. While the aforesaid example coated article of 10/400,080 is excellent in most respects, there is room for improvement.

[0005] In particular, the aforesaid coating when used in applications such as laminated windshields may be problematic in the following areas. First, durability is sometimes lacking. For example, when exposed to conventional brush testing for durability following heat treatment (HT), delamination often occurs. Second, it tends to sometimes be too red/purple with respect to reflective color. It would be desirable if more neutral reflective color could be achieved.

[0006] It would be desirable if one or both of the aforesaid two problems could be addressed and/or solved.

[0007] Turning to another separate and distinct issue, it is known that if one wants to improve a coating's solar performance (e.g., infrared reflection), the thickness of the silver layer(s) can be increased in order to decrease the coating's sheet resistance. Thus, if one wants to improve a coating's solar performance by increasing its ability to reflect infrared (IR) rays or the like, one typically would increase the thickness of the IR reflecting silver layer(s). Unfortunately, increasing the thickness of the silver layer(s) causes visible transmission (T_Y or T_{vis}) to drop. Accordingly, in the past, when one sought to improve the solar performance of a coating in such a manner, it was at the expense of visible transmission. In other words, when solar performance was improved, visible transmission was sacrificed and decreased. Stated another way, it has been difficult to increase the ratio of visible transmission to sheet resistance (i.e., T_{vis}/R_s), especially if heat treatability and/or durability are to be provided. This is why many coatings that block (reflect and/or absorb) much IR radiation have a rather low visible transmission.

[0008] An excellent way to characterize a coated article's ability to both allow high visible transmission and achieve good solar performance (e.g., IR reflection and/or absorption) is the coating's T_{vis}/R_s ratio. The higher the T_{vis}/R_s ratio, the better the combination of the coating's ability to both provide high visible transmission and achieve good solar performance.

[0009] Certain heat treatable coatings cannot achieve very high T_{vis}/R_s ratios, thereby illustrating that their combined characteristic of visible transmission relative to sheet resistance (and solar performance) may be improved. In the past, it has been theoretically possible to increase the T_{vis}/R_s ratio, but not in a commercially

acceptable manner. As an example, Example 2 of U.S. Patent No. 4,786,783 could not survive heat treatment (e.g., heat bending, tempering, and/or heat strengthening) in a commercially acceptable manner. If Example 2 of the '783 Patent was heat treated, the sheet resistance would effectively disappear because the silver layer(s) would heavily oxidize and be essentially destroyed, thereby leading to unacceptable optical properties such as massive (very high) haze, very large ΔE^* values, and unacceptable coloration. For example, because Example 2 of the '783 Patent does not use sufficient layer(s) to protect the silver during HT, the coated article would have very high ΔE^* values (glass side reflective and transmissive) due to heat treatment; ΔE^* over 10.0 and likely approaching 20.0 or more (for a detailed discussion on the meaning of ΔE^* , see U.S. Patent Nos. 6,495,263 and/or 6,475,626, both of which are hereby incorporated herein by reference).

[0010] In view of the above, it may be desirable to improve the aforesaid example coating of 10/400,080 or other coatings at least with respect to durability and/or reflective coloration. Moreover, in certain example instances, it may be desirable to provide a coating with good visible transmission without sacrificing solar performance and/or good solar performance without sacrificing visible transmission. With respect to this latter point, it may in certain example non-limiting situations be desirable for the T_{vis}/R_s ratio to be high in a coating that may be heat treated in a commercially acceptable manner.

BRIEF SUMMARY OF EXAMPLE EMBODIMENTS OF THE INVENTION

[0011] In certain example embodiments of this invention, a layer comprising tin oxide is provided between a layer comprising titanium oxide and a layer comprising silicon nitride. Surprisingly, it has been found that the provision of such a tin oxide inclusive layer between layers comprising silicon nitride and titanium oxide can significantly improve durability of the resulting coated article, especially after heat treatment (HT). The overall adhesion in the coating improves due to this layer comprising tin oxide.

[0012] With respect to durability, the reason for the remarkable improvement in durability is not clear. However, it is believed that it may be related to an

interfacial adhesive problem between silicon nitride and titanium oxide. Undesirable stresses at this interface can be removed by inserting a layer of or including tin oxide between the titanium oxide inclusive layer and the silicon nitride inclusive layer, thereby allowing durability to be significantly improved.

[0013] In certain example embodiments of this invention, there is provided a heat treatable coated article having a visible transmission (T_Y , L_{ta} or T_{vis}) to sheet resistance (R_s) ratio (i.e., a ratio T_{vis}/R_s) of at least 25 after heat treatment (HT), more preferably of at least 30 after HT, even more preferably of at least 32 after HT, still more preferably of at least 34 after HT, and most preferably of at least 36 after HT.

[0014] In other example embodiments of this invention, it has surprisingly and unexpectedly been found that coated articles according to certain example embodiments of this invention have improved heat treatability compared to other known coatings. For example, it has surprisingly been found that coated articles according to certain example embodiments of this invention can be exposed to about 650 degrees C of heat treatment for 12 minutes without suffering a loss in visible transmission of more than 2%. In other words, when the coated article is exposed to about 650 degrees C of heat treatment for 12 minutes, the coated article retains at least 98% of its pre-heat-treatment visible transmission.

[0015] In certain example embodiments, there is provided coated article including a multi-layer coating supported by a glass substrate, the multi-layer coating comprising, from the glass substrate outwardly: a layer comprising titanium oxide; a layer comprising tin oxide located over and contacting the layer comprising titanium oxide; a layer comprising silicon nitride located over and contacting the layer comprising tin oxide; a contact layer; a layer comprising silver located over and contacting the contact layer; a dielectric layer comprising a metal oxide; another layer comprising silver; and another dielectric layer. Other layers may also be provided. The coated article may or may not be heat treated.

[0016] In other example embodiments of this invention, there is provided a heat treatable coated article including a multi-layer coating supported by a glass substrate, the multi-layer coating comprising: a layer comprising tin oxide located directly between and contacting each of a layer comprising titanium oxide and a layer

comprising silicon nitride; at least one layer comprising silver located over the layer comprising tin oxide; and when the coated article is exposed to about 650 degrees C of heat treatment for 12 minutes as a reference, the coated article retains at least 98% of its pre-heat-treatment visible transmission.

BRIEF DESCRIPTION OF THE DRAWINGS

[0017] FIGURE 1 is a cross sectional view of a monolithic coated article according to an example embodiment of this invention – the materials and specific layers thereof in this drawing are provided for purposes of example only.

[0018] FIGURE 2 is a cross sectional view of the coated article of Fig. 1 being used in an IG window unit according to an example embodiment of this invention.

[0019] FIGURE 3 is a cross sectional view of the coated article of Fig. 1 being used in a laminated vehicle windshield according to an example embodiment of this invention.

DETAILED DESCRIPTION OF EXAMPLE EMBODIMENTS OF THE INVENTION

[0020] Referring now more particularly to the accompanying drawings in which like reference numerals indicate like parts throughout the several views.

[0021] Coated articles herein may be used in applications such as vehicle windows (e.g., laminated heat treated windshields), monolithic windows, IG window units, and/or any other suitable application that includes single or multiple glass substrates.

[0022] According to certain example embodiments of this invention, a layer of or including tin oxide (e.g., SnO_2 or $\text{SnO}_2\text{:N}$) is added to the aforesaid example coating of related 10/400,080, between the titanium oxide layer and bottom silicon nitride inclusive layer. Surprisingly, and unexpectedly, it has been found that removal of the titanium oxide/silicon nitride interface, and replacement thereof by a layer comprising tin oxide, allows mechanical durability to be significantly improved especially after heat treatment (HT).

[0023] While the reason for the remarkable improvement in durability is not clear, it is believed that it may be related to an interfacial adhesive problem between silicon nitride and titanium oxide in the aforesaid coating of 10/400,800. In particular, it is believed that internal stress(es) within the titanium oxide layer and silicon nitride layer of the aforesaid coating of 10/400,080 become problematic, especially after heat treatment (HT). These stresses, after HT, seem to cause delamination of the coating upon exposure to conventional brush testing thereby illustrating a lack of durability. It has been found that when a layer of or including tin oxide is provided between the titanium oxide and silicon nitride inclusive layers, the problematic interface between the titanium oxide and silicon nitride is no longer present and a significant improvement in durability results. It has also been found that coated articles according to certain example embodiments of this invention have superior thermal stability upon HT, although in different embodiments of this invention coated articles may or may not be heat treated.

[0024] In certain example embodiments of this invention, there is provided a heat treatable coated article having a visible transmission (T_Y , L_{ta} or T_{vis}) to sheet resistance (R_s) ratio (i.e., a ratio T_{vis}/R_s) of at least 25 after heat treatment (HT), more preferably of at least 30 after HT, even more preferably of at least 32 after HT, still more preferably of at least 34 after HT, and most preferably of at least 36 after HT. The higher this ratio, the better the coated article's combined functionality of providing for both good solar performance (e.g., ability to reflect and/or absorb IR radiation) and high visible transmission. In other words, high visible transmission can be achieved without sacrificing solar performance. In certain example embodiments, coupled with one or more of the aforesaid ratios T_{vis}/R_s , the coated article is heat treatable in a commercially acceptable manner in that: (i) its ΔE^* value (glass side reflective and/or transmissive) due to HT is no greater than about 8.0, more preferably no greater than about 5.0, even more preferably no greater than about 4.0, even more preferably no greater than about 3.0, and sometimes no greater than 2.5; and/or (ii) the coating includes at least one metal inclusive contact layer (e.g., Ni, NiCr, Cr, Ti, TiO, NiCrO_x, ZnO, ZnAlO, Nb, mixtures thereof, or the like) between an IR reflecting layer (e.g., silver layer) and a dielectric layer so as to protect the IR reflecting layer during HT. Additionally, in certain example embodiments of this invention, coated

articles may have a ratio T_{vis}/R_s of at 20 before HT, more preferably of at least 22 before HT, even more preferably of at least 25 before HT, and most preferably of at least 28 before HT. It is noted that in certain embodiments of this invention, coated articles need not have any of the aforesaid T_{vis}/R_s ratios (although they are present in many preferred embodiments). It is further noted that parameters such as R_s , T_{vis}/R_s and ΔE^* are typically measured before lamination in laminated coated article embodiments.

[0025] Factors which may contribute to the surprisingly high T_{vis}/R_s ratios herein include one or more of: example layer stack portions described herein, example layer stoichiometries herein, example layer thicknesses herein, the use of a Si-rich silicon nitride inclusive layer under at least one IR reflecting layer herein, and/or the deposition of different zinc oxide inclusive layers at different oxygen partial pressures. For example, in certain example embodiments, it has surprisingly been found that the use of a combination of a Si-rich silicon nitride inclusive layer and a zinc oxide inclusive layer (e.g., ZnO, ZnAlO_x, or ZnO containing other element(s)) under a layer comprising silver allows the silver be deposited (e.g., via sputtering or the like) in a manner which causes its sheet resistance to be lessened compared to if other material(s) were under the silver. In certain example embodiments, a surprising finding is that the Si-rich silicon nitride inclusive (Si_xN_y) layer(s) allows the ratio T_{vis}/R_s to be increased significantly after HT (e.g., heat strengthening, thermal tempering, and/or heat bending). While it is not certain why this Si-rich layer Si_xN_y works in such a manner, it is believed that the presence of free Si in the Si-rich silicon nitride inclusive layer may allow many atoms such as sodium (Na) which migrate outwardly from the glass during HT to be stopped by the Si-rich silicon nitride inclusive layer before they can reach the silver and damage the same. Thus, it is believed that the oxidation caused by heat treatment allows visible transmission to increase, and that the Si-rich Si_xN_y layer(s) reduces the amount of damage done to the silver layer(s) during HT thereby allowing sheet resistance (R_s) to decrease in a desirable manner.

[0026] Figure 1 is a side cross sectional view of a coated article according to an example non-limiting embodiment of this invention. The coated article includes

substrate 1 (e.g., clear, green, bronze, or blue-green glass substrate from about 1.0 to 10.0 mm thick, more preferably from about 1.0 mm to 3.5 mm thick), and multi-layer coating (or layer system) 27 provided on the substrate 1 either directly or indirectly. The coating (or layer system) 27 may include: dielectric layer 2 of or including titanium oxide, tin oxide inclusive layer 3 which may or may not be doped with nitrogen, dielectric silicon nitride layer 4 which may be Si_3N_4 , or a Si-rich type (which may or may not be doped with Al, stainless steel, or the like), first lower contact layer 7 which contacts and protects IR reflecting layer 9, first conductive and potentially metallic infrared (IR) reflecting layer 9, first upper contact layer 11 which contacts and protects IR reflecting layer 9, dielectric layer 13, another silicon nitride inclusive layer 14 (stoichiometric type or Si-rich type), second lower contact layer 17 which contacts and protects IR reflecting layer 19, second upper contact layer 21 which contacts and protects upper IR reflecting layer 19, dielectric layer 23, and finally protective dielectric layer 25. The "contact" layers 7, 11, 17 and 21 each contact at least one IR reflecting layer (e.g., Ag layer). The aforesaid layers 4-25 make up low-E (i.e., low-emissivity) coating 27 which is provided on glass or plastic substrate 1.

[0027] Compared to the aforesaid coated article of 10/400,080, it has unexpectedly been found that the provision of the tin oxide inclusive layer 3 between the silicon nitride layer 4 and titanium oxide layer 2 is advantageous for the reasons discussed above. Tin oxide layer 3 may or may not be doped with nitrogen (N) (e.g., from about 0-50 atomic %, more preferably from about 1-50 atomic %, even more preferably from about 2-25 atomic %, and most preferably from about 3-20 atomic %) so as to form $\text{SnO}_x\text{:N}$ where x is from about 1.5 to 2.5, more preferably from about 1.8 to 2.2, and may be 2.0 for example. While it is possible that the nitrogen in the tin oxide layer may form an oxynitride in certain example instances, it is believed that the nitrogen therein more likely is trapped in the layer. It is believed that the inclusion of nitrogen in the tin oxide layer may help keep the target clean and/or reduce the effect of internal stresses in the layer so as to allow better heat treatability.

[0028] Titanium oxide layer 2 may or may not be stoichiometric in certain example embodiments of this invention. In other words, for example, the titanium

oxide in layer 2 may be represented by TiO_x in certain example instances, where x is from about 1.5 to 2.5, more preferably from about 1.8 to 2.2. It is possible for the titanium oxide to be doped with other material(s).

[0029] This silicon nitride layer 4 may be doped with Al, B, stainless steel, or the like in different embodiments of this invention (e.g., from 0-15%, more preferably from 1-15%, and most preferably from 5-15%). Moreover, silicon nitride inclusive layer 4 (and/or 14) may be Si-rich in certain embodiments of this invention. In certain example embodiments, it has surprisingly been found that the use of a combination of a Si-rich silicon nitride inclusive layer (4 and/or 14) and a zinc oxide inclusive layer (e.g., ZnO or ZnAlO_x) (7 and/or 17) under a layer comprising silver (9 and/or 19) allows the silver to be deposited (e.g., via sputtering or the like) in a manner which causes its sheet resistance to be lessened compared to if certain other material(s) were under the silver. In certain example embodiments, a surprising finding is that the Si-rich silicon nitride inclusive (Si_xN_y) layer(s) allows the ratio T_{vis}/R_s to be increased significantly after HT (e.g., heat strengthening, thermal tempering, and/or heat bending). It is believed that the presence of free Si in the Si-rich silicon nitride inclusive layer(s) (4 and/or 14) may allow certain atoms such as sodium (Na) which migrate outwardly from the glass 1 during HT to be stopped by the Si-rich silicon nitride inclusive layer before they can reach the silver and damage the same.

[0030] In certain example embodiments, one or both of the potentially Si-rich silicon nitride layers 4 and/or 14 may be characterized by Si_xN_y layer(s), where x/y may be from 0.76 to 1.5, more preferably from 0.8 to 1.4, still more preferably from 0.80 to 1.0 (higher x/y ratios may cause haze levels to increase undesirably). Moreover, in certain example embodiments where little or no oxygen is present in the layer(s) 4 and/or 14, before and/or after HT the Si-rich Si_xN_y layer(s) (4 and/or 14) may have an index of refraction "n" of at least 2.05, more preferably of at least 2.07, even more preferably at least 2.10, and most preferably from about 2.15 to 2.25 (e.g., about 2.20) (e.g., 632 nm) (note: stoichiometric Si_3N_4 has an index "n" of 2.04). Also, the Si-rich Si_xN_y layer(s) (4 and/or 14) in certain example embodiments may have an extinction coefficient "k" of at least 0.001, more preferably of at least 0.003 (note: stoichiometric Si_3N_4 has an extinction coefficient "k" of effectively 0). In

certain example embodiments, the Si-rich Si_xN_y layer(s) 4 and/or 14 has an extinction coefficient "k" of from about 0.01 to 0.02.

[0031] Infrared (IR) reflecting layers 9 and 19 are preferably metallic and/or conductive, and may comprise or consist essentially of silver (Ag), gold, or any other suitable IR reflecting material. These IR reflecting layers help allow coating 27 to have low-E and/or good solar control characteristics. The IR reflecting layer(s) may be slightly oxidized in certain embodiments of this invention.

[0032] The upper contact layers 11 and 21 may be of or include nickel (Ni) oxide, chromium oxide, or a nickel alloy oxide such as nickel chrome oxide (NiCrO_x), or other suitable material(s), in certain example embodiments of this invention. The use of, for example, NiCrO_x for these layers allows durability to be improved. The NiCrO_x layers may be fully oxidized in certain embodiments of this invention (i.e., fully stoichiometric), or may be at least about 50% oxidized in other embodiments of this invention. In certain preferred embodiments, the NiCrO_x for/in these layer(s) may be slightly under-stoichiometric (oxygen deficient). While NiCrO_x is a preferred material for these contact layers, those skilled in the art will recognize that other materials may instead be used. Contact layers 11 and/or 21 (e.g., of or including NiCrO_x) may or may not be oxidation graded in different embodiments of this invention. Oxidation grading means that the degree of oxidation in the layer(s) changes throughout at least part of the thickness of the layer(s) so that for example a contact layer may be graded so as to be less oxidized at the contact interface with the immediately adjacent IR reflecting layer than at a portion of the contact layer(s) further or more/most distant from the immediately adjacent IR reflecting layer. Descriptions of various types of oxidation graded contact layers 11 and 21 are set forth in U.S. Patent No. 6,576,349, the disclosure of which is hereby incorporated herein by reference.

[0033] Lower contact layers 7 and 17 may comprise zinc oxide (e.g., ZnO) in certain example embodiments of this invention. The zinc oxide may contain Al (e.g., to form ZnAlO) or other element(s) in certain example embodiments. In certain alternative embodiments of this invention, another layer (e.g., an oxide of NiCr, an

oxide of Ni, or the like) may be provided between the zinc oxide inclusive layer 7 (or 17) and the nearest IR reflecting layer 9 (or 19).

[0034] Dielectric layer 13 acts as a coupling layer between the two halves of the coating 27, and is of or includes tin oxide in certain embodiments of this invention. However, other dielectric materials may instead be used for layer 13.

[0035] Dielectric layers 23 and 25 may allow the environmental resistance of the coating 27 to be improved, and are also provided for color purposes. In certain example embodiments, dielectric layer 23 may be of or include tin oxide (e.g., SnO_2), although other materials may instead be used. Dielectric overcoat layer 25 may be of or include silicon nitride (e.g., Si_3N_4) in certain embodiments of this invention, although other materials may instead be used such as titanium dioxide, silicon oxynitride, tin oxide, zinc oxide, niobium oxide, or the like such as dielectrics with an index of refraction "n" from 1.6 to 3.0. Layer 23 (and/or other layers in Fig. 1) may be omitted in certain example embodiments of this invention.

[0036] Other layer(s) may also be provided. Moreover, certain layers of coating 27 may be removed in certain embodiments, while others may be added between the various layers or the various layer(s) may be split with other layer(s) added between the split sections in other embodiments of this invention without departing from the overall spirit of certain embodiments of this invention.

[0037] Figure 2 illustrates the coating or layer system 27 being utilized on surface #2 of an IG window unit. Coatings 27 according to any embodiment herein may be used in IG units as shown in Fig. 2. In order to differentiate the "inside" of the IG unit from its "outside", the sun 29 is schematically presented on the outside. The IG unit includes outside glass pane or sheet (i.e., substrate 1 from Fig. 1) and inside glass pane or sheet 31. These two glass substrates (e.g. float glass 1-10 mm thick) are sealed at their peripheral edges by a conventional sealant and/or spacer(s) 33 and may be provided with a conventional desiccant strip (not shown). The panes may then be retained in a conventional window or door retaining frame. By sealing the peripheral edges of the glass sheets and optionally replacing the air in insulating space (or chamber) 30 with a gas such as argon, a typical, high insulating value IG unit is formed. Optionally, insulating space 30 may be at a pressure less than

atmospheric pressure in certain alternative embodiments (with or without a gas in space 30), although this of course is not necessary in all embodiments. While the inner side of substrate 1 is provided with coating 27 in Fig. 2, this invention is not so limited (e.g., coating 27 may instead be provided on the interior surface of substrate 31 in other embodiments of this invention).

[0038] Fig. 3 illustrates the coating of Fig. 1 being used in the context of a laminated window such as a vehicle windshield including heat treated and bent glass substrates 1 and 31. In Fig. 3, the coating 27 is provided on the #3 surface, although this invention is not so limited. In other instances, the coating may be provided on the interior surface of the other substrate.

[0039] Turning back to Fig. 1, while various thicknesses may be used in different embodiments of this invention, example thicknesses and materials for the respective layers on the glass substrate 1 in the Fig. 1 embodiment are as follows, from the glass substrate outwardly:

Table 2 (Example Materials/Thicknesses; Fig. 1 Embodiment)

Layer	Preferred Range (Å)	More Preferred (Å)	Example (Å)
TiO _x (layer 2)	10-1,000 Å	20-100 Å	40 Å
SnO ₂ :N (layer 3)	10-1,000 Å	20-100 Å	40 Å
Si _x N _y (layer 4)	50-450 Å	50-180 Å	120 Å
ZnO _x (layer 7)	10-300 Å	40-150 Å	100 Å
Ag (layer 9)	50-250 Å	80-120 Å	95 Å
NiCrO _x (layer 11)	10-100 Å	20-50 Å	37 Å
SnO ₂ (layer 13)	0-1,000 Å	350-800 Å	550 Å
Si _x N _y (layer 14)	50-450 Å	90-200 Å	115 Å
ZnO _x (layer 17)	10-300 Å	40-150 Å	100 Å
Ag (layer 19)	50-250 Å	80-220 Å	95 Å

NiCrO _x (layer 21)	10-100 Å	20-50 Å	37 Å
SnO ₂ (layer 23)	0-750 Å	70-200 Å	105 Å
Si ₃ N ₄ (layer 25)	0-750 Å	120-320 Å	185 Å

[0040] In certain example embodiments of this invention, the total thickness of the combination of layers 2, 3, 4 and 7 is substantially less than the total thickness of the combination of layers 13, 14 and 17. In certain example embodiments, the combination of layers 2, 3, 4 and 7 is no more than about 50% as thick as the combination of layers 13, 14 and 17, more preferably no more than about 45% as thick, and most preferably no more than about 41% as thick (e.g., in the above example column, the combination of layers 2, 3, 4 and 7 is about 39.2% as thick as the combination of layers 13, 14 and 17). Moreover, in certain example embodiments of this invention, the combination of layers 2, 3, 4 and 7 is no more than about 400 Å thick, more preferably no more than about 350 Å thick, and most preferably no more than about 300 Å thick. It has surprisingly been found that by following the aforesaid ranges, reflective color of the resulting coated article (e.g., windshield) can be kept from becoming too red/purple.

[0041] In certain example embodiments of this invention, coated articles herein may have the following low-E (low emissivity) characteristics set forth in Table 3 when measured monolithically (before any optional HT). The sheet resistances (R_s) herein take into account all IR reflecting layers (e.g., silver layers 9, 19) in the coating, unless expressly stated to the contrary.

Table 3: Low-E/Solar Characteristics (Monolithic; pre-HT)

Characteristic	General	More Preferred	Most Preferred
R _s (ohms/sq.):	<= 5.0	<= 3.5	<= 3.0
E _n :	<= 0.07	<= 0.04	<= 0.03
T _{vis} /R _s :	>= 20	>= 22	>= 25

[0042] In certain example embodiments, coated articles herein may have the following characteristics, measured monolithically for example, after HT:

Table 4: Low-E/Solar Characteristics (Monolithic; post-HT)

Characteristic	General	More Preferred	Most Preferred
R_s (ohms/sq.):	≤ 4.5	≤ 3.0	≤ 2.5
E_n :	≤ 0.07	≤ 0.04	≤ 0.03
T_{vis}/R_s :	≥ 25	≥ 30	≥ 32 (or ≥ 34 or 36)

[0043] As explained above, the rather high values of the ratio T_{vis}/R_s in Tables 3 and 4 are indicative of an excellent combination of high visible transmission and good solar performance (e.g., IR reflection).

[0044] Moreover, coated articles including coatings 27 according to certain example embodiments of this invention have the following optical characteristics (e.g., when the coating(s) is provided on a clear soda lime silica glass substrate 1 from 1 to 10 mm thick) (HT or non-HT). In Table 5, all parameters are measured monolithically, unless stated to the contrary. In Table 5 below, R_gY is visible reflection from the glass (g) side of the monolithic article, while R_fY is visible reflection from the side of the monolithic article on which coating/film (f) (i.e., coating 27) is located. It is noted that the ΔE^* values are of course due to HT (e.g., heat strengthening, tempering, and/or heat bending). All Ill. D65 parameters are at 10 degrees.

Table 5: Optical Characteristics (Mono-HT)

Characteristic	General	More Preferred
T_{vis} (or TY) (Ill. D65):	$\geq 70\%$	$\geq 75\%$
a^*_t (Ill. D65):	-4.0 to +1.0	-3.0 to 0.0
b^*_t (Ill. D65):	-3.0 to +4.0	-1.0 to 3.0
L^*_t (Ill. D65):	85 to 95	88 to 93
R_gY (Ill. D65):	1 to 10%	3 to 9%
a^*_g (Ill. D65):	-4.5 to +2.0	-3.0 to 0.0
b^*_g (Ill. D65):	-5.0 to +10.0	-4.0 to +9.0

L^*_g (Ill. D65):	25 to 40	28 to 34
R_fY (Ill. D65):	1 to 10%	1 to 8%
a^*_f (Ill. D65):	-14.0 to 10.0	-11.0 to 3.0
b^*_f (Ill. D65):	-9.0 to 20.0	-7.0 to 12.0
L^*_f (Ill. D65):	25 to 35	28 to 32
ΔE^*_t (transmissive):	≤ 8.0	$\leq 5.0, 4.0, 3.0$ or 2.5
ΔE^*_g (glass side reflective):	≤ 8.0	$\leq 5.0, 4.0, 3.0$ or 2.5
Transmissive Haze:	≤ 0.4	≤ 0.35 , or ≤ 0.25
R_s (ohms/square):	≤ 5.0	≤ 3.0 , or ≤ 2.5

[0045] However, it is noted that in other example embodiments, the visible transmission may be lower (e.g., as low as 60%, or even 40—50% in certain instances).

[0046] The value(s) ΔE^* is important in determining whether or not there is matchability, or substantial color matchability upon HT, in the context of certain embodiments of this invention. Color herein is described by reference to the conventional a^* , b^* values. The term Δa^* is simply indicative of how much color value a^* changes due to HT (the same applies to Δb^*). If color changes too much upon HT (e.g., if ΔE^* is over 10), then the product may not be commercially acceptable. A very high value of ΔE^* may also be indicated of destruction of the Ag layer during HT, and/or of massive haze.

[0047] The term ΔE^* (and ΔE) is well understood in the art and is reported, along with various techniques for determining it, in ASTM 2244-93 as well as being reported in Hunter et. al., *The Measurement of Appearance*, 2nd Ed. Cptr. 9, page 162 et seq. (John Wiley & Sons, 1987). As used in the art, ΔE^* (and ΔE) is a way of adequately expressing the change (or lack thereof) in reflectance and/or transmittance (and thus color appearance, as well) in an article after or due to HT. ΔE may be calculated by the "ab" technique, or by the Hunter technique (designated by employing a subscript "H"). ΔE corresponds to the Hunter Lab L, a, b scale (or L_h , a_h , b_h). Similarly, ΔE^* corresponds to the CIE LAB Scale L^* , a^* , b^* . Both are deemed

useful, and equivalent for the purposes of this invention. For example, as reported in Hunter et. al. referenced above, the rectangular coordinate/scale technique (CIE LAB 1976) known as the L*, a*, b* scale may be used, wherein:

L* is (CIE 1976) lightness units

a* is (CIE 1976) red-green units

b* is (CIE 1976) yellow-blue units

and the distance ΔE^* between L*_o a*_o b*_o and L*₁ a*₁ b*₁ is:

$$\Delta E^* = \{(\Delta L^*)^2 + (\Delta a^*)^2 + (\Delta b^*)^2\}^{1/2} \quad (1)$$

where:

$$\Delta L^* = L^*_1 - L^*_o \quad (2)$$

$$\Delta a^* = a^*_1 - a^*_o \quad (3)$$

$$\Delta b^* = b^*_1 - b^*_o \quad (4)$$

where the subscript "o" represents the coating (or coated article) before heat treatment and the subscript "1" represents the coating (coated article) after heat treatment; and the numbers employed (e.g., a*, b*, L*) are those calculated by the aforesaid (CIE LAB 1976) L*, a*, b* coordinate technique. In a similar manner, ΔE may be calculated using equation (1) by replacing a*, b*, L* with Hunter Lab values a_h, b_h, L_h. Also within the scope of this invention and the quantification of ΔE^* are the equivalent numbers if converted to those calculated by any other technique employing the same concept of ΔE^* as defined above.

[0048] As explained above, coated articles according to certain example embodiments of this invention may have a ΔE^* value (glass side reflective and/or transmissive) due to HT which is no greater than about 8.0, more preferably no greater than about 5.0, even more preferably no greater than about 4.0, even more preferably no greater than about 3.0, and sometimes no greater than about 2.5. These values are indicative of commercially acceptable heat treatable coated articles. In laminated embodiments herein, the ΔE^* value is typically measured before lamination.

[0049] Moreover, coated articles including coatings 27 according to certain example embodiments of this invention have the following characteristics (e.g., when the coating(s) is provided on surface #3 (or other suitable surface) of a heat treated laminated vehicle windshield as shown in Fig. 3). In certain examples, the glass substrates in this respect may be clear glass substrates about 2.1 mm thick and the PVB interlayer may be about 0.76 mm thick. Again, all Ill. D65 measurements were at 10 degrees. Here, since the coating is on surface #3 as shown in Fig. 3, the glass side measurements would be taken from the side of the laminate which would be the interior of the vehicle.

Table 6: Characteristics (Laminated Windshield – HT)

Characteristic	General	More Preferred
L _{ta} (Ill. A, 2 deg.):	$\geq 70\%$	$\geq 75\%$
T _{vis} (or TY) (Ill. D65):	$\geq 70\%$	$\geq 75\%$
a* _t (Ill. D65):	-5.0 to +1.0	-3.5 to 0.0
b* _t (Ill. D65):	-3.0 to +5.0	-1.0 to 3.5
L* _t (Ill. D65):	85 to 95	88 to 93
R _{gY} (Ill. D65):	1 to 12%	3 to 10%
a* _g (Ill. D65):	-4.5 to +2.0	-3.0 to 0.0
b* _g (Ill. D65):	-12.0 to +10.0	-10.0 to +8.0
L* _g (Ill. D65):	30 to 40	34 to 38
R _{fY} (Ill. D65):	1 to 11%	1 to 10%
a* _f (Ill. D65):	-10.0 to 5.0	-5.0 to 3.0
b* _f (Ill. D65):	-10.0 to 20.0	-8.0 to 12.0
L* _f (Ill. D65):	30 to 40	34 to 37
Transmissive Haze:	≤ 0.4	≤ 0.35 , or ≤ 0.33
R _{solar} (ISO 9050):	≥ 25	≥ 30
T _{solar} (TS):	≤ 47	≤ 45

EXAMPLE 1

[0050] The following Example 1 is provided for purposes of example only, and is not intended to be limiting. The following examples included the listed layer stack set forth below, from the clear glass substrate outwardly.

Table 7: Layer Stack for Examples

Layer	Thickness
Glass Substrate	about 2 to 2.3 mm
TiO _x	40 Å
SnO ₂ :N	40 Å
Si _x N _y	120 Å
ZnAlO _x	100 Å
Ag	95 Å
NiCrO _x	37 Å
SnO ₂	550 Å
Si _x N _y	115 Å
ZnAlO _x	100 Å
Ag	95 Å
NiCrO _x	37 Å
SnO ₂	105 Å
Si ₃ N ₄	185 Å

[0051] Moreover, as explained herein, the bottom two silicon nitride layers (Si_xN_y) (which are doped with Al in this Example) are preferably non-stoichiometric and Si-rich. As explained above, it has been found that the use of a combination of a Si-rich silicon nitride inclusive layer and a zinc oxide inclusive layer (e.g., ZnAlO_x) under a layer comprising silver allows the silver to be deposited in a manner which causes its sheet resistance to be lessened (which is desirable) compared to if other material(s) were under the silver. In certain example embodiments, the Si-rich silicon

nitride inclusive (Si_xN_y) layer(s) allows the ratio T_{vis}/R_s to be increased significantly after HT.

[0052] The sputtering process used in forming the coated article of Example 1 is set forth below, with a line speed of 5.0 m/min in a Terra-G flat glass coater. Below, Ar, O and N refer to respective gas flows of these gases in the coater in units of sccm. Pressure is in units of hPa. Volts refers to volts at the cathode. The silver and NiCr targets were planar targets, the Sn targets were Twin-Mags, the ZnAl targets were twin-PLs, the Ti target was a ceramic Ti DCM target, and the SiAl targets were DCMs. Power is indicative of the power set point. For cathodes K11 and K12, the ratio of reactive gas flow to power was about 4.43 ml/kW. Moreover, λ settings for the coater were as follows: K15 – 174; K28 – 184; and K29 – 181.

Table 8: Sputter Processing - Example 1

Cathode	Target	Power(kW)	Ar	O	N	Pressure	Volts
K05	Ti	15	350	25	0	1.88E-03	242.5
K09	Sn	7	250	380	50	2.31E-03	251.8
K11	SiAl	40	350	0	277	2.71E-03	242.5
K12	SiAl	40	350	0	277	1.75E-03	251.8
K15	ZnAl	51.3	250	525	0	1.99E-03	279.9
K21	Ag	8.0	150	0	0	1.27E-03	388.6
K25	NiCr	18.3	250	150	0	1.44E-03	510.1
K28	Sn	29.6	250	491	350	2.23E-03	260.7
K29	Sn	31.2	250	501	350	2.21E-03	257.2
K39	Sn	33.5	250	533	350	2.27E-03	268.2
K40	Sn	32.4	250	503	350	2.36E-03	256.7
K41	Sn	32.2	250	523	350	2.53E-03	267.0
K43	SiAl	65.0	350	0	371	2.05E-03	291.5
K46	ZnAl	43.5	250	545	0	1.49E-03	238.5

K49-b	Ag	9.5	150	0	0	1.32E-03	474.1
K50	NiCr	17.5	250	150	0	1.32E-03	498.7
K54	Sn	46.2	250	653	350	2.12E-03	311.0
K59	SiAl	69.0	350	0	514	2.78E-03	293.7
K60	SiAl	69.0	350	0	514	3.12E-03	336.0

[0053] After being sputter deposited onto the glass substrate, Example 1 had the following example characteristics after being subjected to HT along with another lite, spaced apart by powder, in a conventional furnace (HT in furnace with eight element heat zones (479, 520, 540, 590, 620, 655, 660, 665 degrees C respectively, line speed of 28 cm/minute):

Table 9: Characteristics of Example 1 (Monolithic - HT)

Characteristic	Example 1
Visible Trans. (Y)(Ill. D65, 10 deg.):	79.58%
a*	-1.12
b*	0.58
L*	91.50
ΔE^*_t	about 4-5
Glass Side Refl. (RY)(Ill D65, 10 deg.):	6.86%
a*	-2.77
b*	7.65
L*	31.48
ΔE^*_{g}	about 2-3
Film Side Refl. (FY)(Ill. D65, 10 deg.):	6.38%
a*	-9.14
b*	9.87

L*	30.36
R _s (ohms/square):	2.5
T _{vis} /R _s (post-HT):	31.83
Transmissive Haze:	0.19

[0054] The coated article of Example 1 was then laminated to another glass sheet (e.g., via PVB or index oil) to form a laminated windshield. The laminated windshield had the following characteristics. The laminate was as shown in Fig. 3, with the coating on the #3 surface (i.e., on the inner glass sheet). The "film side" measure measurements in Table 10 are as viewed from what would be the vehicle exterior, whereas the "glass side" measurements are as viewed from what would be the vehicle interior, since the coating is on the #3 surface of the laminate in this example.

Table 10: Characteristics of Example 1 (Laminated - HT)

Characteristic	Example 1
L _{ta} (T _{vis}) (Ill. A, 2 deg.)	76.0%
Visible Trans. (Y)(Ill. D65, 10 deg.):	76.15%
a*	-2.33
b*	2.63
L*	89.93
Glass Side Refl. (R _Y)(Ill D65, 10 deg.):	9.02%
a*	-2.60
b*	-5.21
L*	36.02
Film Side Refl. (F _Y)(Ill. D65, 10 deg.):	8.63%
a*	-2.40
b*	-6.30

L*	35.26
Transmissive Haze:	0.31
R _{solar} (ISO 9050):	27.5
T _{solar} :	45.0

[0055] In addition to the advantages set forth above, it has surprisingly and unexpectedly been found that coated articles according to certain example embodiments of this invention have improved heat treatability compared to other known coatings. For example, it has surprisingly been found that coated articles according to certain example embodiments of this invention can be exposed to about 650 degrees C of heat treatment in a box furnace for 12 minutes without suffering a loss in visible transmission of more than 2% or delamination. In other words, when the coated article (typically in monolithic form) is exposed to about 650 degrees C of heat treatment for 12 minutes, the coated article retains at least 98% of its pre-heat-treatment visible transmission. This temperature and time are used as a reference(s) herein, and are not intended to be limiting; these values being used to illustrate the improved heat treatability (or improved thermal stability) of certain example embodiments of this invention.

[0056] As an example of heat treatment, the coated article may be heat treated at the same time as another glass lite, with the two lites being spaced from one another in the furnace by a known powder material. During such heat treatment, the coated article may be located in the furnace so as to be spaced from another glass lite via a known powder such as Separol. Thus, it can be seen that thermal stability can be significantly improved according to certain example embodiments of this invention.

[0057] Certain terms are prevalently used in the glass coating art, particularly when defining the properties and solar management characteristics of coated glass. Such terms are used herein in accordance with their well known meaning (unless expressly stated to the contrary) – see parent serial number 10/400,080, incorporated herein by reference. For example, the terms "heat treatment" and "heat treating" as used herein mean heating the article to a temperature sufficient to achieve thermal

tempering, bending, and/or heat strengthening of the glass inclusive article. This definition includes, for example, heating a coated article in an oven or furnace at a temperature of least about 580 or 600 degrees C for a sufficient period to allow tempering, bending, and/or heat strengthening, and also includes the aforesaid test for thermal stability at about 650 degrees C. In some instances, the HT may be for at least about 4 or 5 minutes, or more.

[0058] Haze is a scattering of energy, a percentage of energy lost as it scatters. Haze may be measured, for example, by a BYK Gardner transmission haze meter as will be appreciated by those skilled in the art.

[0059] While the invention has been described in connection with what is presently considered to be the most practical and preferred embodiment, it is to be understood that the invention is not to be limited to the disclosed embodiment, but on the contrary, is intended to cover various modifications and equivalent arrangements included within the spirit and scope of the appended claims. For example, and without limitation, single silver stacks may be used in certain example embodiments of this invention, as opposed to the illustrated double silver stacks.